



# Raman spectroscopy study of the influence of additives (Cr-, Cr/Al-, and Gd) on $\text{UO}_2$ dissolution behavior

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## Abstract

In this work, Raman spectroscopy (RS) has been used to study the behavior of spent nuclear fuel (SNF) under disposal conditions. In particular, evolutionary Accident-Tolerant Fuels (ATFs) consisting in  $\text{UO}_2$  doped with Cr and Cr/Al; as well as neutron absorbers ( $\text{UO}_2$  doped with Gd), has been manufactured and studied. The prepared pellets have been exposed to three different pH leachants (inert media, carbonated media, and young cement water with calcium) for distinct times. RS analyses have been conducted in both the leached and the unleached pellets. The results show that the addition of Cr, and Cr/Al does not lead to a noteworthy change of the pellet surface under the three media considered. Gd doping induces a higher intensity of the so-called “defects band,” which is not dramatically affected after leaching. No secondary phases have been observed. These results are in good agreement with previous studies by XRD and SEM, thus validating RS as an accurate analytical technique.

## Introduction

Raman spectroscopy (RS) is commonly known as a molecule identification and quantification technique with many outstanding advantages, such as no sample modification, no need of sample preparation, and low amount of sample required [1]. More importantly for the purpose of this study, RS has greatly evolved since its discovery in 1928 [2], making it possible nowadays to use this technique not only in the laboratory with sophisticated high-precision equipment, but also to develop field equipment for in situ analysis and on line monitoring of different processes. Moreover, this technique, envisioned as a stand-off technique (Raman spectra acquired where the spectrometer, and therefore the operator, is at some distance from the sample under analysis [3]), allows the measurement of dangerous samples, such as explosives or nuclear fuel, while maintaining safety criteria, of a paramount importance for the operator's safety [4].

In the nuclear field, RS has been used for studying the behavior of Spent Nuclear Fuel (SNF) under dry conditions. In these studies, SNF surrogates (both  $\text{UO}_2$  [5, 6] and doped  $\text{UO}_2$  [7, 8]) have been analyzed by RS, in oxidizing environments, mimicking some operations at the dry storage facility. Under these conditions, the SNF matrix (*i.e.*,  $\text{UO}_2$ ) could undergo oxidation and form a number of uranium oxides, to finally reach  $\text{U}_3\text{O}_8$ , a phase with important implications in the fuel integrity [9]. The obtained spectral fingerprints of these oxides was used to identify them by in situ experiments performed at different conditions of temperature and oxygen availability [6]. For doing so, dedicated measurement protocols were optimized and successfully applied in the study of irradiated fuel in hot cells [10].

Beyond dry storage, it is accepted that after a number of stages in the back-end of the nuclear fuel cycle, SNF will be stored in a deep geological repository (DGR) [11]. The main concern in this repository is the potential release of radionuclides when groundwater contact the SNF. Once uranium reaches its solubility limit in groundwater, it can form uranyl secondary phases. Some of those phases may retain the released radionuclides, and retard their mobility and migration to the biosphere [11–13]. Although Raman spectra of the different possible secondary phases are known both experimentally [14] and simulated [15], the use of RS to analyze fuel dissolution and formation of these phases

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is scarce in the literature and limited to  $\text{UO}_2$  [16] or MOX fuel [17].

In this work, we propose going one-step further, using RS to study the dissolution behavior of Accident-Tolerant Fuels (ATFs). ATFs have emerged as a promising advancement to improve the safety and performance of nuclear reactors. Designed to withstand higher temperatures and to reduce the release of radioactive fission products, ATFs are engineered to have enhanced material properties compared to conventional  $\text{UO}_2$  fuel. One of the shortest-term concepts is doping conventional  $\text{UO}_2$  fuels with different elements. These additive-based  $\text{UO}_2$  fuels incorporate small quantities of metal oxides, primarily chromia ( $\text{Cr}_2\text{O}_3$ ) or chromia-alumina ( $\text{Cr}_2\text{O}_3\text{-Al}_2\text{O}_3$ ) [18–20]. Besides ATFs, one widely researched type of doped nuclear fuel includes  $\text{UO}_2$  doped with small amounts of gadolinium oxide ( $\text{Gd}_2\text{O}_3$ ) because it acts as a Burnable Neutron Absorber (BNA) that helps managing reactivity within the reactor core [21].

The behavior of SNF in DGR environments is influenced by the chemical and physical properties of groundwater, including pH, complexing agents, and redox conditions. The  $\text{UO}_2$  typically undergoes oxidative dissolution when exposed to oxygenated water or some anionic species that promote complexation (*e.g.*, bicarbonates) [22]. The presence of certain dopants may affect the solubility, surface reactivity, and the formation of secondary phases during the corrosion process [23]. Preliminary data on doped  $\text{UO}_2$  suggests that its corrosion behavior under repository conditions is similar to undoped  $\text{UO}_2$  at neutral and basic pH levels, but differences occur in high-carbonate environments or under mildly oxidizing conditions [24]. These variations are associated to the dopants that alter surface properties, influencing the dissolution rates that influence overall fuel stability. By understanding how dopants affects the behavior of  $\text{UO}_2$  under repository conditions, experiments on the stability of SNF should be refined, supporting in the design of safer and more efficient nuclear waste management strategies. One approach of refining these experiments is to monitor in situ the behavior of these fuels under controlled conditions approximating DGR environments, such as groundwater compositions. Prior to this type of in situ studies, here we examine the surface alteration of doped  $\text{UO}_2$  fuels by analyzing leached doped fuel pellets using RS, and comparing the Raman data obtained with the same unleached pellets. In addition to Raman investigations, the fuel samples studied here have been also analyzed by X-Ray Diffraction (XRD) and Scanning Electron Microscope (SEM), both prior and after the same leaching experiments. The results of such study can be found in a recently published study [25].

## Materials and methods

Samples studied in this work consisted in doped pellets:  $\text{UO}_2\text{-0.06 wt% Cr}_2\text{O}_3$ ,  $\text{UO}_2\text{-0.05 wt% Cr}_2\text{O}_3\text{-0.02 wt% Al}_2\text{O}_3$ , and  $\text{UO}_2\text{-4.5 wt% Gd}_2\text{O}_3$ . The materials were obtained by following a conventional solid-state sintering process.  $\text{UO}_2$  (provided by ENUSA) and high-purity dopants ( $\text{Cr}_2\text{O}_3$ ,  $\text{Al}_2\text{O}_3$ , and  $\text{Gd}_2\text{O}_3$ , provided by Alfa-Aesar) were mixed in weighted proportions to obtain the desired stoichiometry. Each powder mix was subjected to mechanical uniaxial pressing at 700 MPa to produce the green pellets. Those were first calcined at three temperatures (100, 300, and 500 °C) and then sintered at 1675 °C in a reducing atmosphere (4.7%  $\text{H}_2/\text{N}_2$ ). Sintered pellets were then polished and finally etched (at 90% of the sintering temperature in 4.7%  $\text{H}_2/\text{N}_2$  atmosphere) to reveal the grain structure. A more detailed description can be found in a previous study [25].

Dissolution tests were performed using three leaching solutions (300 mL): (a) inert media (labeled as PC, 20 mM  $\text{NaClO}_4$ , pH = 7.2); (b) carbonated media (identified as BC, 19:1 mM of  $\text{NaHCO}_3\text{:NaCl}$ , pH = 8.9); and (c) young cement water with calcium (named YCW, pH ~ 13.5 and simulating hyperalkaline conditions with calcium). Static dissolution tests were performed in an Ar glove box using autoclaves under 4.7% $\text{H}_2\text{-N}_2$  gas ( $p_{\text{H}_2} = 0.37 \pm 1$  bar) at  $22 \pm 3$  °C. Each doped  $\text{UO}_2$  sample was exposed to leachants for 136 days (PC), 168 days (BC), and 379 days (YCW), respectively. Details of the leachant preparation in the laboratory and leaching experiment results are given elsewhere [25].

Both, the as-prepared sintered and the post-leached pellets were analyzed in detail by RS (Horiba LabRam HR evolution spectrometer, Jobin Yvon Technology, using a He–Ne laser with an excitation wavelength of 632.8 nm). Each spectra represents the averaged spectra from ten individual spectrums recorded at random locations of each disk. Raman data analysis has consisted in (1) calculating the exact position of the Raman features by using the second derivative method [5], and (2) using these positions as input parameters in the profile analysis, performing the fitting of the data with Lorentzian functions and calculating the intensity of each band. Uncertainty of the band's positions has been calculated as the standard deviation of the 10 individual spectra. In the case of the intensity ratios, the uncertainty analysis has been done by error propagation from the uncertainty obtained in every individual fitting.

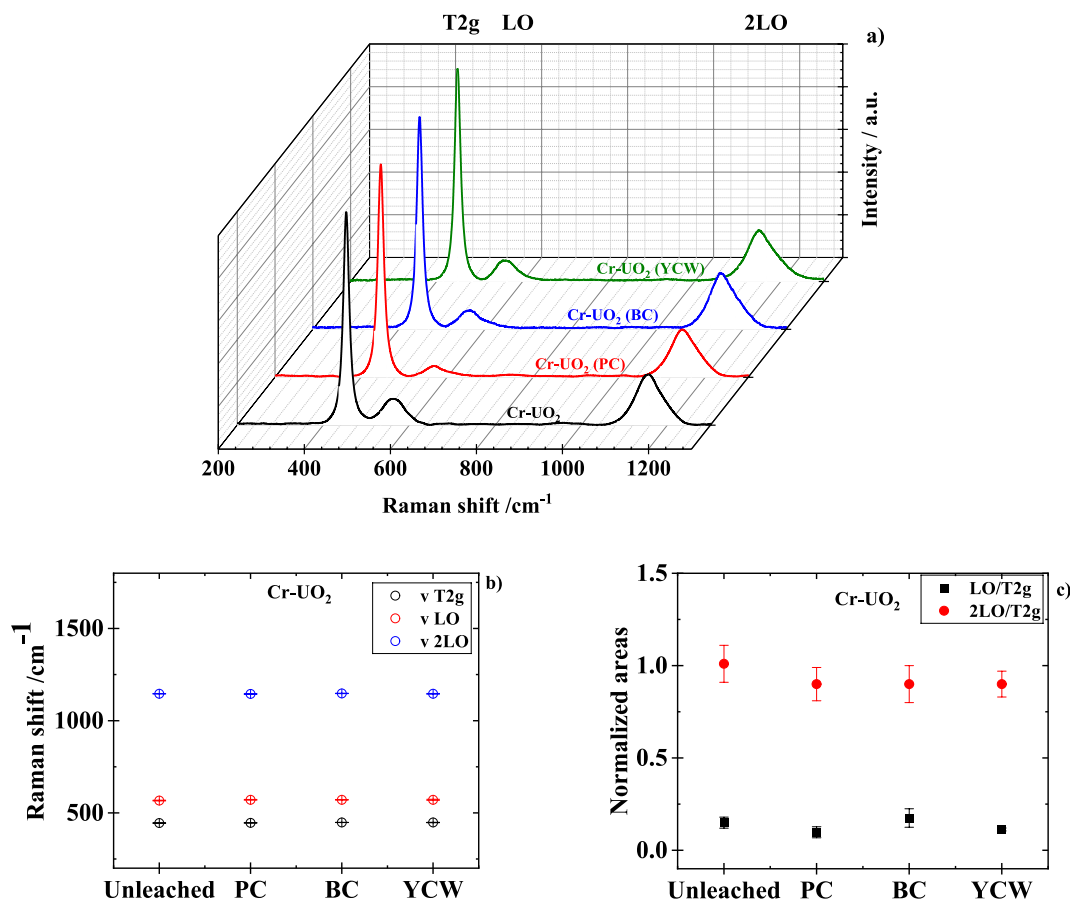
## Results and discussion

Figures 1, 2 and 3 show the averaged Raman spectra acquired from the surface of the unleached pellets of Cr-, Cr/Al-, and Gd-doped  $\text{UO}_2$ , respectively, and after being

immersed in PC, BC, and YCW. The Raman spectra of the as-prepared materials (Figs. 1a, 2a, 3a) show the typical  $\text{UO}_2$  Raman spectrum, containing some features consistent with the literature [5, 8, 26]: (a)  $445\text{ cm}^{-1}$  ( $T_{2g}$ ), corresponding to the vibration of the U–O bond within the  $\text{UO}_2$  matrix; (b)  $575\text{ cm}^{-1}$  (LO phonon) associated with the presence of defects in the fluorite lattice; and (c)  $\sim 1140\text{ cm}^{-1}$  (2LO), representing the first overtone of this phonon vibration. In the case of  $\text{UO}_2$  matrices doped with trivalent elements, as in this work, the substitution of U(IV) by M(III) ( $M = \text{Cr, Al or Gd}$ ) creates a charge imbalance in the lattice. On the one hand, this imbalance is compensated by the creation of oxygen vacancies, as recently demonstrated by Murphy et al. [27] studying Cr-doped  $\text{UO}_2$  fuel by Electron Paramagnetic Resonance. On the other hand, the compensation proceeds through the creation of U(V), as proposed by Kim et al. [28] in their study about the air-oxidation of Gd-doped  $\text{UO}_2$  and recently confirmed by Vinograd et al. [29]. In any case, Raman spectroscopy is sensitive to the creation of these defects in the lattice, by the appearance of a broad band around  $540\text{ cm}^{-1}$  in the spectra [7, 30]. Often, both this band at  $540\text{ cm}^{-1}$  and the

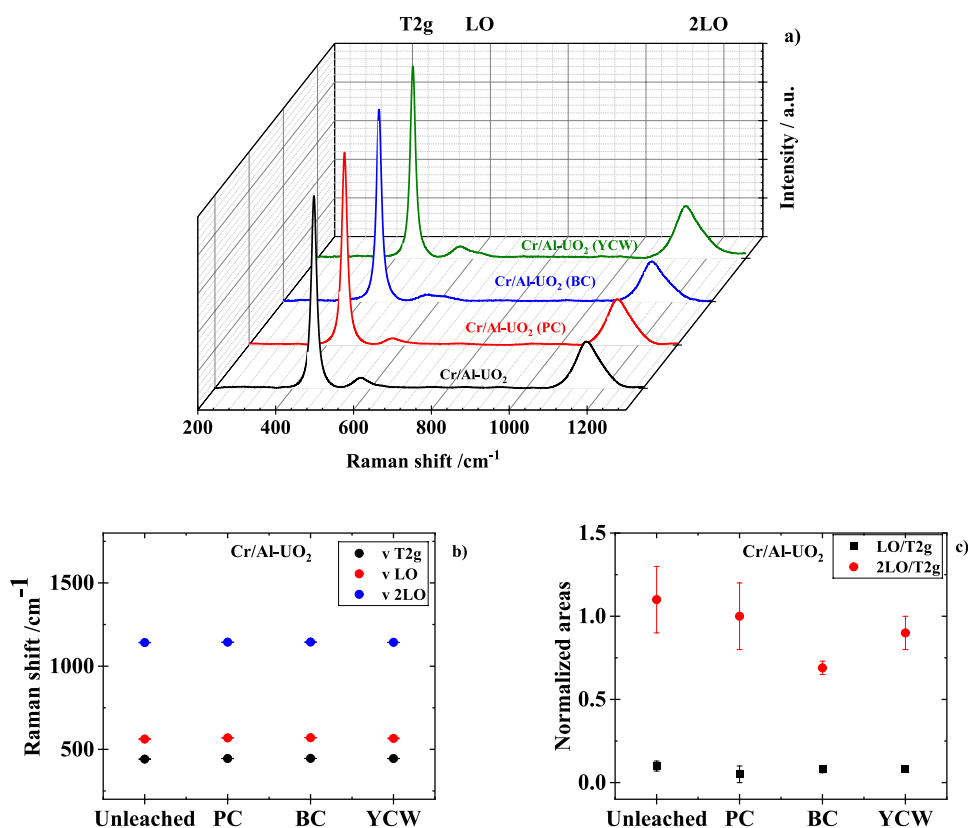
LO phonon at  $575\text{ cm}^{-1}$  overlap, giving place to a broad band generally known as “defects band” [31]. For the sake of simplicity, this feature is named as “LO” in ulterior analyses and plots. These modes give a very good estimation of the sensitivity of RS to the amount of dopant that has entered into the  $\text{UO}_2$  matrix. In Figs. 1a, 2a, the intensity of this band is quite low (with intensity values around 0.2, Figs. 1c and 2c). However, in the case of Gd-doped  $\text{UO}_2$ , it can be seen that the band is much more intense (Fig. 3a), with an intensity value of around 3.7 (Fig. 3c). This behavior is explained by the higher solubility of Gd in the solid solution with  $\text{UO}_2$ , compared to Cr and Al [7].

Regarding the Raman spectra of the leached samples, they are also shown in Figs. 1 to 3. As a general comment, no uranyl secondary phases were identified on the surface for the three doped  $\text{UO}_2$  pellets. The presence of secondary phases, such as studtite ( $\text{UO}_4 \cdot 2\text{H}_2\text{O}$ ) or schoepite ( $(\text{UO}_2)(\text{OH})_2 \cdot x\text{H}_2\text{O}$ ) on the surface of  $\text{UO}_2$  leads to additional Raman bands in the region of  $700\text{--}900\text{ cm}^{-1}$  [32], which are not observed here. In the following paragraphs, a detailed description of the behavior of each one of the



**Fig. 1** a Comparison of the Raman spectra on Cr-doped  $\text{UO}_2$  pellets: unleached (black) and after immersed in PC (red), BC (blue), and YCW (green) solutions; b position of  $T_{2g}$ , LO, and 2LO as a function of leachant; and c peak area ratio of  $\text{LO}/T_{2g}$  and  $2\text{LO}/T_{2g}$

**Fig. 2** **a** Comparison of the Raman spectra on Cr/Al-doped  $\text{UO}_2$  pellets: unleached (black) and after immersed in PC (red), BC (blue), and YCW (green) solutions; **b** position of  $T_{2g}$ , LO, and 2LO as a function of leachant; and **c** peak area ratio of  $\text{LO}/T_{2g}$  and  $2\text{LO}/T_{2g}$



Raman modes described before for every doped material studied is presented.

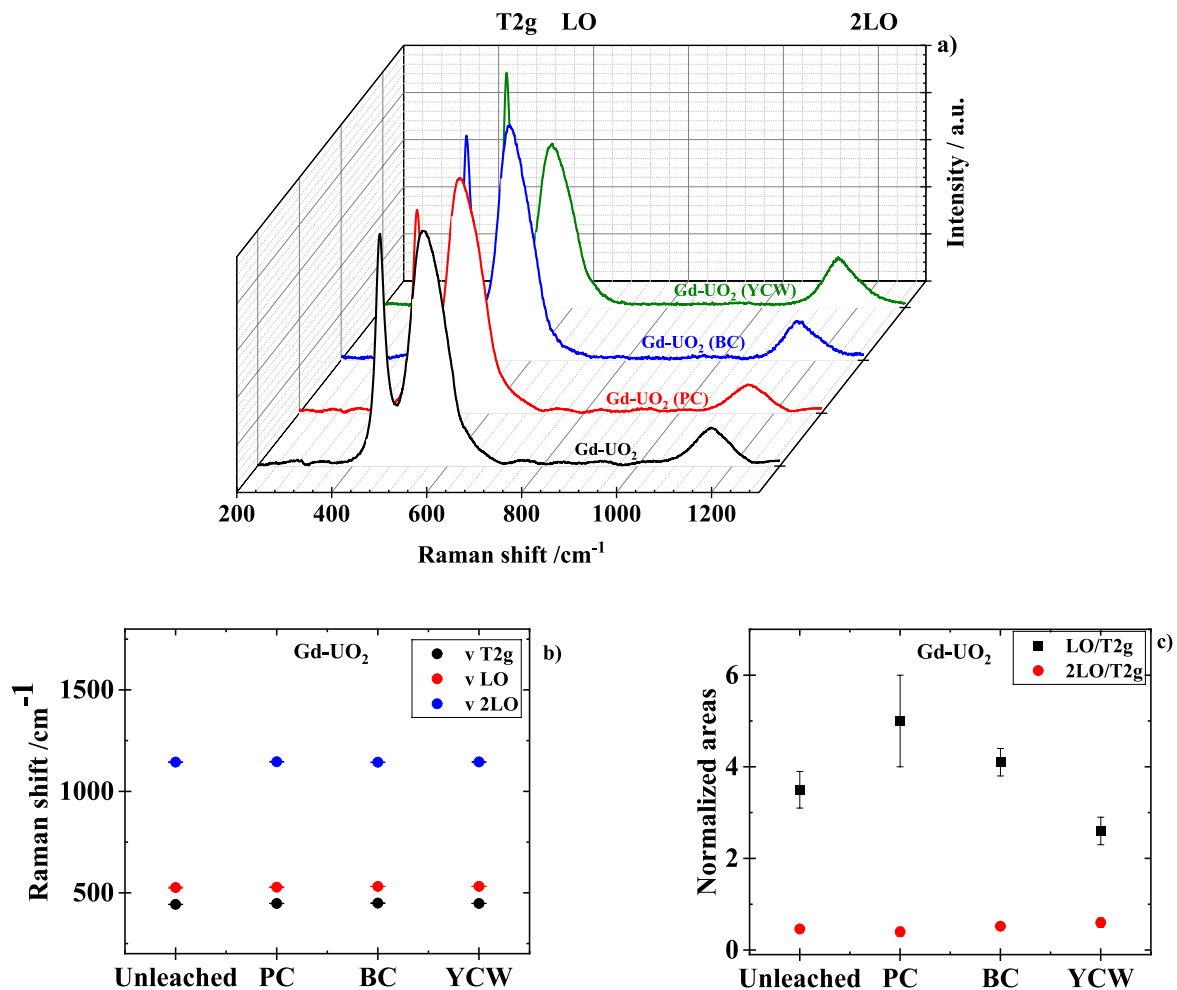
From the analysis of Cr- $\text{UO}_2$  (Fig. 1b), it can be concluded that there is no significant changes from the three main band position, *i.e.*,  $T_{2g}$ , LO, and 2LO. The calculated areas of LO and 2LO bands normalized to that of the  $T_{2g}$  band are plotted against the leachant in Fig. 1c. The data of the  $2\text{LO}/T_{2g}$  area ratios for leached pellets seems constant within error margins, but slightly decreased compared to the one of the unleached pellet, which may imply an increase in defects in the surface oxide. The change in the peak area ratios of  $2\text{LO}/T_{2g}$  should instead be attributed to the chemical behavior of surface U atoms [33] as the dissolution process results in the formation of vacancies within the crystal structure.

The same analytical procedure was applied for the Cr/Al- $\text{UO}_2$  sample. Fig. 2a shows the Raman spectrum of Cr/Al- $\text{UO}_2$  prior and post-leached at pH 7.2, 8.9, and 13.5, where one can see the previously mentioned main features of  $\text{UO}_2$ . Fig. 2b shows the position the three main bands with no appreciable changes among unleached and leached Cr/Al- $\text{UO}_2$  samples. However, the position of the  $T_{2g}$  band is slightly deviated at lower frequencies ( $441\text{ cm}^{-1}$ ) compared to the sample where Cr is the solely dopant ( $445\text{ cm}^{-1}$ ). The normalized area of the LO and 2LO bands to  $T_{2g}$  (Fig. 2c) for leached sample in BC decreased compared to the pre-tested

sample. The most notably drop corresponds to the  $2\text{LO}/T_{2g}$  peaks ratio for sample leached in BC.

The Raman spectra of Gd- $\text{UO}_2$  significantly changes (Fig. 3a) compared to the previous untreated samples. While the relative intensity of the two bands at  $445\text{ cm}^{-1}$  and  $1150\text{ cm}^{-1}$  decreases, the broad band between  $500$  and  $700\text{ cm}^{-1}$  increases. Consequently, the  $2\text{LO}/T_{2g}$  ratio decreases (Fig. 3c) compared to Cr- and Cr/Al- $\text{UO}_2$  (Figs. 1c and 2c, respectively) possibly indicating more distortion for Gd, but the LO intensity increases with the doping level, as in this sample with 4.5%. This can be clearly observed when compared the  $\text{LO}/T_{2g}$  normalized area in Fig. 3c, with Figs. 1c and 2c for the as-sintered samples with lower amount of trivalent dopants. For this dopant, while the normalized area of the 2LO band remains almost the same, the area ratio of the LO band significantly increases at pH 7.2 and 8.9. On the contrary, it decreases at pH 13.5 (Fig. 3c) together with a broadening of the LO peak maybe caused by some surface oxidation. Anyhow, this potential oxidation, if any, should be very small, since there are no clear hints of the presence of the band related to the excess of oxygen in the lattice at around  $630\text{ cm}^{-1}$  [5].

All the mentioned results from RS were also confirmed by XRD analysis on the leached surfaces [25]. These primary data from ex situ leached samples allow us continuing with in situ and/or *operando* experiments in order to provide



**Fig. 3** **a** Comparison of the Raman spectra on Gd-doped UO<sub>2</sub> pellets: unleached (black) and after immersed in PC (red), BC (blue), and YCW (green) solutions; **b** position of T<sub>2g</sub>, LO, and 2LO as a function of leachant; and **c** peak area ratio of LO/T<sub>2g</sub> and 2LO/T<sub>2g</sub>

additional insights into the nuclear fuel behavior and the effect of dopants under different environments.

## Conclusion

In this work we have demonstrated the capacity of Raman spectroscopy in the analysis of the dissolution of new fuels, such as ATFs (Cr- and Cr/Al-doped UO<sub>2</sub>) and BNA (Gd-doped UO<sub>2</sub>), in three different leachants at pH 7.2, 8.9, and 13.5, relevant in the context of SNF behavior in a deep geological repository. The results, in agreement with the ones obtained by XRD show that the addition of Cr, and Cr/Al does not lead to a noteworthy change of the pellet surface under the three media considered. The incorporation of 4.5 wt% of Gd<sub>2</sub>O<sub>3</sub> doping induces a marked increase in the intensity of the so-called “defect band” (mainly attributed to the LO phonon associated with lattice disorder), because of defects formation caused by doping.

The analyses of the leached samples surfaces by RS show the absence of precipitated secondary phases containing uranium. It can be inferred that RS is a suitable technique in the analysis of this type of experiments (prior and after the leaching test). The results obtained here open the door to explore the kinetic corrosion of doped UO<sub>2</sub> in ground-water using in situ conditions and RS as the monitoring technique.

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**Author contributions** AMP and LJB performed/analyzed the Raman spectra, and wrote the original draft. NRV conceived of the idea, performed the dissolution experiments and reviewed the work. JC and HG reviewed the findings of this work.



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**Data availability** Data sets generated during the current study are available from the corresponding author on reasonable request.

## Declarations

**Competing interests** The authors have no conflicts of interest to declare.

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